In situ reduction and oxidation of nickel from solid oxide fuel cells in a Titan ETEM

A. Faes^{1,2}, Q. Jeangros¹, J.B. Wagner³, A. Hessler-Wyser¹, J. Van herle², R. Dunin-Borkowski³

- 1. Interdisciplinary Centre for Electron Microscopy (CIME), Ecole Polytechnique Fédérale de Lausanne (EPFL), CH-1015 Lausanne, Switzerland
- Laboratory of Industrial Energy Systems (LENI), EPFL, CH-1015 Lausanne, Switzerland
 Center for Electron Nanoscopy (CEN), Technical University of Denmark (DTU), DK-2800 Lyngby, Denmark

antonin.faes@epfl.ch

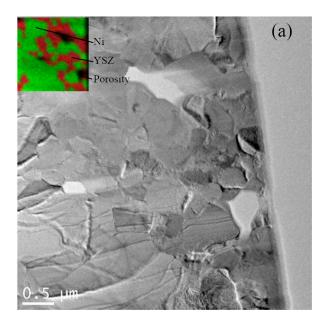
Keywords: In situ ETEM, nickel oxide, reduction, RedOx, SOFC

Solid Oxide Fuel Cells (SOFC) common technology is based on anode-supported cells composed of nickel-yttria stabilized zirconia (Ni-YSZ) cermet. The nickel is in oxide state (NiO) during SOFC production and is reduced to metallic nickel during the first operation. The microstructure influences the SOFC electrochemical performance [1] as well as its stability for long-term use [2]. Oxidation of the nickel catalyst can occur at high fuel utilization and due to air leakage. The volume change from Ni to NiO can be detrimental for the thin supported electrolyte [3].

In situ transmission electron microscopy enables to acquire further knowledge on the mechanisms behind the reduction and oxidation of nickel in the Ni-YSZ SOFC anode.

The in situ reduction and re-oxidation of the FIB prepared TEM lamellae is performed in a FEI Titan equipped with an environmental cell. Figure 1 presents a bright field micrograph of the anode microstructure before and during in situ reduction. The reduction started at 400°C under 1.4 mbar of hydrogen. The volume contraction due to NiO reduction is compensated by formation of nanoporosity in the Ni grain. Nanoporosity was previously observed by Waldbillig et al. during ex situ reduction of Ni-YSZ composite TEM lamellae [4].

- 1. S. C. Singhal, K. Kendall, High Temperature Solid Oxide Fuel Cell Fundamentals, Design and Applications. (Elsevier, 2003).
- 2. D. Simwonis, F. Tietz, D. Stoever, Solid State Ionics 132, (2000) p241.
- 3. A. Faes, A. Nakajo, A. Hessler-Wyser, D. Dubois, S. Modena, A. Brisse, J. Van herle,
- J. Power Sources (2009) in press.
- 4. D. Waldbillig, A. Wood, D. G. Ivey, J. Power Sources 145, (2005) p206.
- 5. This research was supported by the European Institute for Energy Research (EIfER), contract n° N43/C06/019.



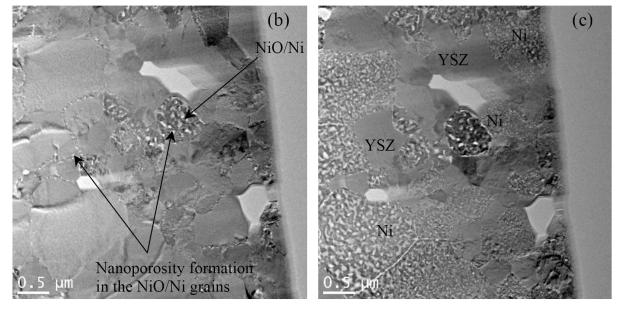


Figure 1. Bright field micrographs of the sample (a) at room temperature before reduction and (b) at 450°C under 1.4 mbar H2 (c) at 500°C under 1.4 mbar H2. The EDX map of the micrograph is on the top left of the image (a). YSZ grains are left unchanged by the reduction while the Ni grains exhibit internal nanoporosity (bright spots on the micrographs taken at 450°C and 500°C).